

## The effect of mechanical activation on the relaxor properties of PbMg<sub>1/3</sub>Nb<sub>2/3</sub>O<sub>3</sub> - PbFe<sub>1/2</sub>Nb<sub>1/2</sub>O<sub>3</sub> solid solution ceramics

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Recently we have found out that the high-energy mechanical activation of the starting oxides reduces dramatically the frequency shift  $\Delta T$  and increases by about 20 K the temperature  $T_m$  of the dielectric permittivity maximum for ceramics of a classical ferroelectric-relaxor PbMg<sub>1/3</sub>Nb<sub>2/3</sub>O<sub>3</sub> (PMN) sintered from the mixture of these oxides [1]. Similar changes of  $\Delta T$  and  $T_m$  were predicted previously, basing on the first-principle calculations, for PMN with an increased degree of a short-range ordering of Mg<sup>2+</sup> and Nb<sup>5+</sup> cations [2]. However, as we used for mechanical activation a high-energy planetary-centrifugal mill AGO-2 with both jars and balls made of the stainless steel, one could expect that a small amount of iron from a milling media incorporates into the processed powder and the (1-x)PbMg<sub>1/3</sub>Nb<sub>2/3</sub>O<sub>3</sub> - xPbFe<sub>1/2</sub>Nb<sub>1/2</sub>O<sub>3</sub> (PMN-xPFN) solid solution is formed during sintering. The formation of such solid solution, in principle, could explain the observed increase of  $T_m$  as well as a smaller unit cell parameter value (4.037 Å instead of  $\approx 4.05$  Å for a usual PMN ceramics) observed for PMN ceramics studied in [1]. The scope of the present work was to obtain several PMN-PFN solid solution compositions by both the usual solid-state synthesis and by high-energy mechanochemical synthesis and compare their relaxor properties and the values of the unit cell parameters.

Energy dispersive X-ray analysis (EDXA) has shown that the PMN samples obtained using high-energy mechanochemical synthesis contain about 1 wt.% of iron. Raman studies revealed that an additional band appears in the spectrum of such ceramics at 700 cm<sup>-1</sup>. Similar band is present in the Raman spectrum of PFN and it corresponds to the Fe-O stretching mode [3]. Thus at least a part of iron incorporates into the lattice of PMN. If one assumes that all this iron incorporates into the crystal lattice of PMN, the 0.8PMN-0.2PFN solid solution composition would be formed, which contains some MgO excess. Our studies have shown that for the PMN-PFN compositions from the 0 ≤ x ≤ 0.2 range both the unit cell parameter and the  $\Delta T$  values are smaller when ceramics is obtained using high-energy mechanochemical synthesis. On the other hand  $T_m$  values were very similar for the ceramics of the same composition obtained by both methods. Thus, one may conclude that the increase of  $T_m$  in the PMN ceramics obtained using high-energy mechanochemical synthesis may be ascribed, at least partially, to the formation of the PMN-PFN solid solution. However the main origin of the unit cell parameter and  $\Delta T$  changes observed for such ceramics seems to be not the incorporation of iron from the milling media into the lattice, but rather the effect of high-energy mechanical activation.

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1. S.I. Raevskaya, A.A. Gusev, V.P. Isupov, et al., *Ferroelectrics* **525**, 37 (2018).
2. S. Prosandeev, L. Bellaiche, *Phys. Rev. B* **94**, 180102 (2016).
3. Yu.I. Yuzyuk, I.P. Raevski, S.I. Raevskaya, et al., *J. Alloys Compds* **695**, 182 (2017).